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Nanostructured Fe-Ni Sulfide: A Multifunctional Material for Energy Generation and Storage

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Abstract: Multifunctional materials for energy conversion and storage could act as a key solution for growing energy needs. In this study, we synthesized nanoflower-shaped iron-nickel sulfide (FeNiS) over a nickel foam (NF) substrate using a facile hydrothermal method. The FeNiS electrode showed a high catalytic performance with a low overpotential value of 246 mV for the oxygen evolution reaction (OER) to achieve a current density of 10 mA/cm², while it required 208 mV at 10 mA/cm² for the hydrogen evolution reaction (HER). The synthesized electrode exhibited a durable performance of up to 2000 cycles in stability and bending tests. The electrolyzer showed a lower cell potential requirement for a FeNiS-Pt/C system (1.54 V) compared to a standard benchmark IrO₂-Pt/C system (1.56 V) to achieve a current density of 10 mA/cm². Furthermore, the FeNiS electrode demonstrated promising charge storage capabilities with a high areal capacitance of 13.2 F/cm². Our results suggest that FeNiS could be used for multifunctional energy applications such as energy generation (OER and HER) and storage (supercapacitor).

Keywords: FeNiS; electrocatalyst; cyclic voltammetry; supercapacitor; cyclic stability; flexibility

1. Introduction

The US Energy Information Administration (EIA) estimates a 28% growth in total world energy consumption from 2015 to 2040 [1]. Increasing global population has resulting in a dramatic demand for energy in household appliances, portable electronic devices, automobiles, aerospace vehicles and industrial equipment, thus creating an urgent need for clean, renewable energy generation for storage devices [2–8]. Fuel cells and supercapacitors are among the most widely used energy conversion and storage devices [9,10]. In order to survive growing needs, multifunctional materials could assist in providing efficient remedial solutions. Thio-spinels are widely used materials for multifunctional energy generation in storage devices [11]. Synergistic chemical composition using different transition metal compounds provides several advantages in terms of tailoring properties and obtaining ultimately



promising performances for a wide range of applications spanning the environment [12], energy [13] and electrocatalysis [14].

Hydrogen is considered to be an efficient clean energy fuel and can be easily be obtained from water electrolysis [15] through a hydrogen evolution reaction (HER) leading to a high energy density in the order of 141.86 MJ/kg [7]. On the other hand, oxygen gas produced from an oxygen evolution reaction (OER) through water splitting could serve as an essential candidate for air batteries, fuel cells, or as an environmentally friendly gas [5,16]. In theory, based on thermodynamic considerations, water splitting should occur at 1.23 V (vs. a reversible hydrogen electrode (RHE)); however, in practice this process requires extra potential called overpotential [17]. To overcome this challenge, an efficient catalyst that reduces the reaction overpotential is needed. Several materials such as IrO₂, RuO_2 and Pt are used as benchmark catalysts for water electrolysis in the open literature [18,19]. However, their industrial applicability is limited due to lower cycle stability and higher cost as they are precious metals-based. Hence, current water catalysis demands a multifunctional catalyst that is environmentally friendly, earth-abundant, highly stable and economically viable to synthesize. Moreover, supercapacitors are among the widely applied energy storage devices most used, along with batteries, due to their high power density, fast charge-discharge rate, safe operation, wide temperature use range and long lifespan [20,21]. Current supercapacitor research focuses on achieving high energy density along with high power density [3,8]. Transition metal oxides and sulfides are well-known energy storage materials due to their lower costs, higher charge storage capacities and wide availability in nature [22–26].

In this study, nanostructured iron-nickel oxide (FeNiO) and iron-nickel sulfide (FeNiS) were synthesized using a facile hydrothermal method. The synthesized electrodes were used for energy generation and storage in alkaline electrolytes. In the electrocatalytic studies, the FeNiS electrode required a low overpotential for OER (246 mV) and HER (208 mV) at 10 mA/cm². Stable performance was observed during both the stability test over 2000 cycles and the bending test of the electrode. The overall water splitting performance was analyzed using a FeNiS-Pt/C system. A lower cell potential of 1.54 V was required for FeNiS-Pt/C, compared with 1.56 V for IrO₂-Pt/C, to achieve the current density of 10 mA/cm². The synthesized FeNiS electrode showed promising charge storage capabilities with a high areal capacitance of 13.2 F/cm² at a scan rate of 1 mV/s and 4.9 F/cm² at a current density of 5 mA/cm². Our study yielded a favorable multifunctional electrode material for energy generation and storage devices.

2. Results and Discussion

Due to the amorphous nature of the synthesized samples, X-ray diffraction (XRD) patterns did not show any diffraction peaks. The morphology of the synthesized electrodes was analyzed using scanning electron microscopy (SEM). Figure 1 shows SEM images for the synthesized FeNiO and FeNiS samples grown over nickel foam (NF). FeNiO formed a structure like fused nanosheet balls on the NF surface, as observed in Figure 1a,b. After sulfurization, the sheet structure became more evident with a flower-shaped morphology of FeNiS nanosheets, as shown in Figure 1c,d. Porous nanosheets directly grown over the NF were anticipated to ease the access to the electrolyte and accelerate the electron transport from the nanosheets to the NF substrate, therefore boosting the electrochemical performance.

The XPS survey spectra of FeNiO and sulfurized FeNiO are given in Figure S1. Figures 2a–c and 2d–f show the high resolution, core level XPS spectra of the FeNiO and FeNiS surfaces, respectively. The substantial range of iron, nickel and mixed Fe/Ni oxide/hydroxide compounds, and the complexity of their photoelectron spectra (including multiplet splitting and satellite features), makes any interpretation of unknown species on both the mixed metal oxide grown from iron and nickel precursor species, and the oxidized metal sulphide, very difficult. Consequently, the XPS spectral interpretation discussed below must be considered within this context, with the interpretation being tentative due the spectral complexity and range of different Fe/Ni oxide/hydroxide- and sulphur-containing species potentially present at the surface. The work of the Surface Science Western group is very useful in understanding

the complex Fe and Ni 2p peak shapes shown in Figure 2 [27–30]. Our XPS Fe 2p and Ni 2p peak fits are indicative, and for a greater understanding of the Fe 2p and Ni 2p peak structure we refer the reader to the work mentioned above [27–30].

Considering first the FeNiO surface, the Fe:Ni ratio determined from the XPS measurements was 1:2.8, so the stoichiometry of this catalyst was Ni_{2.8}FeO_x. Stoichiometries of Fe:Ni between 1:2 and 1:2.6 have been reported by other authors for $FeNiO_x$ when using a precursors molar ratio of 1:1 [31], so our value of 1:2.8 was not dissimilar from previous work. The binding energy of the Ni $2p_{3/2}$ and Ni $2p_{1/2}$ peaks are observed at 856.2 and 873.5 eV, respectively (Figure 2b). The peak fit shows that although the Ni 2p_{3/2} maximum peak intensity arises at 856.2 eV, there are clearly two components contributing to this peak, with the presence of a lower binding energy component at 854.7 eV in addition to a higher binding energy component. A narrow linewidth peak for this lower binding energy component is indicative of Ni^{2+} oxide species [27,28]. There is also a strong broad satellite peak at 861.8 eV and two small satellite components at higher binding energies. The component binding energy shifts, FWHMs and general peak fit are in excellent agreement with that of NiFe₂O₄, given by Biersinger et al. [29], the only deviation being the intensity of the 856.2 eV component, which is higher in Figure 2b compared to the standard NiFe₂O₄ [29]. This deviation may have arisen from the variation in the composition of this catalyst compared with the NiFe₂O₄ standard, or from the presence of other Ni²⁺/Ni³⁺ species present at the surface such as OH⁻ [27]. The presence of hydroxide species at the surface can be seen by the higher binding energy OH⁻ component at 531.6 eV in addition to oxide, O^{2-} , at 530.0 eV in the O 1 s spectrum (Figure 2c).

The binding energy of the Fe $2p_{3/2}$ and Fe $2p_{1/2}$ peaks (Figure 2a) occur at 711.7 and 724.7 eV, respectively. Such binding energies indicate that the iron was in an Fe³⁺ oxidation state. However, compared to the Ni 2p spectrum, the peak positions and general peak shape of the Fe 2p spectrum are not so consistent with the spectrum for NiFe₂O₄ given by Biersinger et al. [29]. In particular, there is increased intensity between the $2p_{3/2}$ and $2p_{1/2}$ peaks for the spectrum shown in Figure 2a compared with standard NiFe₂O₄. This increased intensity can be attributed to the presence of the Ni L₃M_{2,3}M_{4,5} (¹P) Auger peak at ~712 eV. There is also a Ni Auger L₃M_{2,3}M_{4,5} (³P) peak at ~706 eV, which is the source of the broad shoulder to the low binding energy side of the Fe $2p_{3/2}$ peak. This is in agreement with the Ni-rich nature of this mixed metal oxide, and the increased spectral intensity between the Fe $2p_{3/2}$ and $2p_{1/2}$ peaks has also been observed for a Ni₆Fe₄O_x catalyst reported by Smith et al. [31]. The presence of the Ni L₃M_{2,3}M_{4,5} Auger peaks make peak-fitting the Fe 2p peak very difficult. There are also Fe³⁺ satellite peaks at around 8.5 eV from the main $2p_{3/2}/2p_{1/2}$ peaks [30].

For the FeNiS surface, the dominant peaks for the Fe 2p (Figure 2d) and Ni 2p (Figure 2e) spectra are again associated with the presence of surface oxides and, as for the FeNiO surfaces, show binding energies of ~712 and ~856 eV for the Fe $2p_{3/2}$ and Ni $2p_{3/2}$ peaks, respectively. Oxidation of metal sulphides when exposed to air or water is well-known [32-35] and result from the higher free energy of formation to form the metal oxides/hydroxides/sulphates. The Fe 2p spectrum shows no clear evidence of an iron sulphide peak at lower binding energies than the oxide/hydroxide peaks. However, the Ni Auger $L_3M_{2,3}M_{4,5}$ (³P) broad shoulder does have a slightly different shape than that of FeNiO, and includes a peak at 707.5 eV that gives rise to a better peak fit. In their work on FeNiS, Piontek et al. [36] observed a small peak at 707.2 eV, which they ascribed to FeS₂. This binding energy is consistent with others reported for FeS₂ [33]. Previous careful XPS analysis by Buckley and Woods [34,35] has shown that the un-oxidized surface of $Fe_{0.89}S$ exhibits a Fe $2p_{3/2}$ binding energy of 707.5–708.0 eV, but oxidation in air and water occurs very rapidly, leading to a shift in the binding energy to 711 eV. Consequently, we consider that there a small iron sulphide peak that was present in our Fe 2p spectrum for FeNiS (Figure 2d) at 707.5 eV. The Ni 2p spectrum (Figure 2e) shows a clearer peak at ~853.0 eV, attributable most probably to NiS [37,38] (NiS₂ is reported to have a slightly higher binding energy of 853.6 eV [39]).

The S 2p spectrum found in Figure 2f is complex and very similar to that observed by Piontek et al. for FeNiS catalysts [36]. There are three different chemical states given in the peak fit, sulphide,

oxysulphide and sulphate. S $2p_{3/2}$ and S $2p_{1/2}$ peaks are given for all the three different chemical states and in each case there are Fe (lower binding energy) and Ni (higher binding energy) peaks for each chemical state (as reported in the literature for Fe/Ni sulphides and Fe/Ni sulphates). Thus, there are 12 peaks included in the peak fit. The broad higher binding energy S 2p peak (comprised of four components, numbered 9-12) centered at ~168.5 eV is easily identified and corresponds to the S $2p_{3/2}/S$ $2p_{1/2}$ peaks for Fe/Ni sulphate species. The higher intensity lower binding energy S 2p peak envelope (components 1–4) is again comprised of two S chemical states, each with a S $2p_{3/2}$ and S $2p_{1/2}$ doublet. Peak fitting shows the binding energies of these two main S 2p_{3/2} components peaks to be 161.3 eV (peak 1) and 162.4 eV (peak 3), which are both generally consistent with the presence of a metal sulphide compound. Piontek et al. [36] proposed that these peaks correspond to oxysulphide species. However, oxidation in water or air has previously shown the formation only of oxide or sulphate species, not oxysulphide [32,33]. Oxysulphide would be expected to occur at a higher binding energy than typical metal sulphides and, as we propose below, such species are only likely to occur as a thin layer at the interface between the sulphide and sulphate, thus giving rise to relatively small intensity peaks. The higher intensity S 2p_{3/2} peak at 162.4 eV is in very good agreement with that of NiS reported by others [38,39]. The lower binding energy S $2p_{3/2}$ peak at 161.2 eV agrees well with the value of 161.1 eV given by Buckley and Woods for $Fe_{0.89}S$ [34,35]. The presence of spectral intensity between the sulphide and sulphate peaks (components 5–8) most probably represents oxysulphide Ni and Fe species at the interface between the sulphide and sulphate, but could also be caused by Ni/Fe-deficient sulphide at the surface of the FeNiS, resulting from oxidation of the metallic species [34].

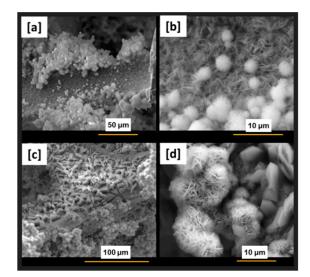


Figure 1. SEM images of (a,b) FeNiO and (c,d) FeNiS.

The presence of a relatively strong FeS peak in the S 2p spectrum, but a very weak peak in the Fe 2p spectrum, would appear to be inconsistent. However, S 2p photoelectrons have higher kinetic energies compared to Fe 2p photoelectrons. Calculations of the difference in electron attenuation length based on the universal equation by Seah [40] showed the S 2p photoelectrons to have an analysis depth 50% greater than the Fe 2p photoelectrons. Consequently, since the sulphide was buried beneath the surface oxide/hydroxide, this difference in analysis depth can explain the absence of the FeS peak in the Fe 2p spectrum but its presence in the S 2p spectrum. However, NiS appears in the Ni 2p spectrum (with the Ni 2p photoelectrons exhibiting a lower kinetic energy than the Fe 2p photoelectrons). This would indicate that the outer surface of the mixed metal sulphide is rich in Ni and depleted in Fe. Furthermore, the results of Piontek et al. also concur with this interpretation [36]. Their XPS Fe, Ni and S spectra are generally very similar to those observed here, apart from the presence of a weak Fe-S peak. However, in the study by Piontek et al., the Ni-S peak was substantially stronger than their weak Fe-S peak, even for samples with an Fe:Ni ratio of 2:1, and for samples where the Fe:Ni ratio was reported to be

1:1 the Ni-S peak was even stronger than the Ni oxide/hydroxide peaks [36]. For our catalysts, the Fe:Ni ratio in the metal sulphide determined from the S 2p peak was 1:1.3, thus we would expect a significantly weaker Fe-S peak compared to that given by Piontek et al., and our NiS peak is of a much lower intensity than the Ni oxide/hydroxide peaks. Thus, there is greater oxidation of our mixed-metal sulphide compared to the catalysts synthesized by Piontek et al. [36], and we would expect a particularly weak FeS peak considering the sulphide to be Ni rich.

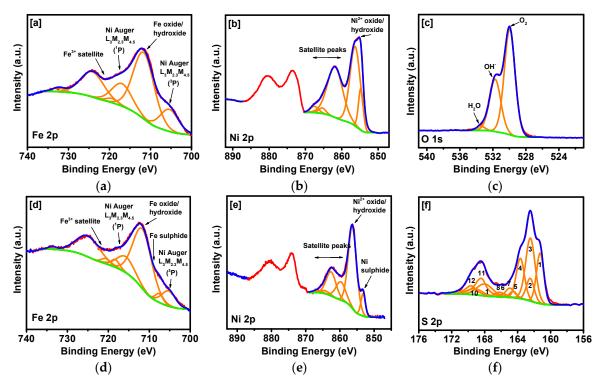


Figure 2. Peak-fitted XPS spectra of (a–c) FeNiO and (d–f) FeNiS.

Linear scan voltammetry (LSV) was used to analyze the water-splitting reaction using synthesized catalytic electrodes. Figure 3a summarizes the catalytic performance of all the catalyst electrodes. It is essential to have a lower overpotential value for better catalytic performance along with a low Tafel slope to ensure better reaction kinetics [41,42]. The lowest overpotential of 246 mV was observed for FeNiS compared to FeNiO (277 mV), IrO₂ (309 mV) and NF (410 mV) at a current density of 10 mA/cm². FeNiS showed comparable reaction kinetics with a Tafel slope of 49 mV/dec compared to the benchmark IrO₂ catalyst shown in Figure 3b. Moreover, our results are comparable to other previously reported studies, as summarized in Table S1 (Supplementary Materials). The better electrocatalytic performance could be further correlated to the active electrochemical surface area of the catalytic electrode. Highly porous nanostructured sheets of NiFeS would allow easy access to the electrolyte and an easy escape of oxygen from the electrode resulting in lower overall series resistance (Figure S2a). The stability of the OER catalyst electrode was studied using a stability test, bending test and chronoamperometry (CA) test. Figure 3c,d shows the excellent stability of the catalytic electrode after the 2000 cycle and bending tests at a 45° angle. Furthermore, the Nyquist plot at 0.5 V shows no significant change in resistance of the overall system, suggesting an excellent catalytic stability of the FeNiS electrode. Moreover, a chronoampere study presented in Figure S2d also showed stable electrode performance with minimal loss in current even after 15 h. Thus, from the overall study, FeNiS shows promising characteristics as an OER catalyst to be used as a low-cost alternative with respect to benchmark catalysts such as IrO2.

6 of 12

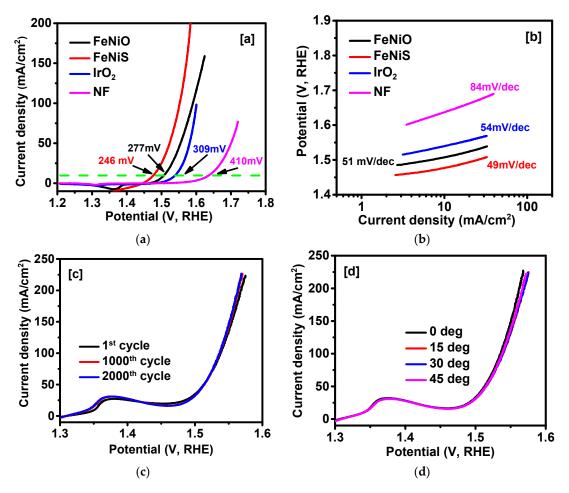


Figure 3. (a) Polarization curves for OER, (b) Tafel slopes for FeNiO, FeNiS, IrO₂ and NF, (c) LSV curves during stability test for OER and (d) LSV curves at different bending angles.

The electrocatalytic properties for the HER reaction were analyzed for all electrodes using the LSV test. It can be understood from Figure 4a that FeNiS shows a low overpotential value of 208 mV, while FeNiO requires an overpotential of 253 mV to achieve a current density of 10 mA/cm². Although FeNiS shows overpotential values comparable to those previously reported, competing with the catalytic properties of the benchmark Pt/C catalyst for HER still remains a challenge (Table S2). Furthermore, the Tafel slopes for FeNiS, FeNiO and Pt/C were observed as 109, 129 and 56 mV/dec, respectively (Figure 4b). As observed in the previous section, catalytic electrodes should perform in a stable manner when used as HER catalyst. Figure 4c clearly shows the LSV curves after the 1st, 1000th and 2000th cycles, with almost no deterioration in catalytic properties. Similar behavior was observed from LSV curves during bending tests at different angles (Figure 4d). The Nyquist plots obtained from electrochemical impedance spectroscopy (EIS) at -1.3 V during both the stability and bending tests are given in Figure S3a,b. The charge transfer resistance showed very small variation upon stability and bending tests, as given in Figure S3c.

An electrolyzer cell was fabricated to study water splitting using a two-electrode system. In the previous three-electrode system, the OER study of FeNiS showed a lower overpotential for IrO_2 , which is one of the standard benchmark catalysts, while in the HER test obtaining an overpotential lower than the benchmark Pt/C catalyst still remains a challenge. From our study, we observed that IrO_2 can only perform well as an OER catalyst and Pt/C as a HER catalyst (Figure S4). Hence, in the two electrode study, we used FeNiS as the OER catalyst along with Pt/C as the HER catalyst (Figure 5). The comparison was made with IrO_2 and Pt as standard catalysts. It was observed that FeNiS-Pt/C

required a lower potential of 1.54 V compared to IrO₂-Pt/C (1.56 V) to achieve a current density of 10 mA/cm^2 . The reported results are lower compared to the majority of previously reported values in alkaline media, as observed in Table S3. Furthermore, FeNiS-Pt/C showed stable performance during continuous water electrolysis for 16 h, as shown in Figure 5b.

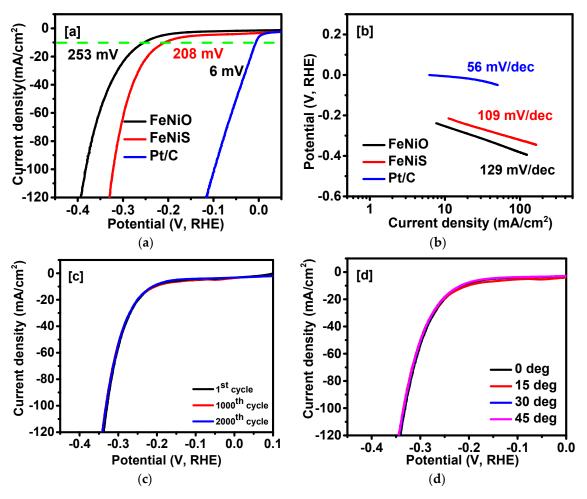


Figure 4. (**a**,**b**) Polarization curves and Tafel slopes for FeNiO, FeNiS and Pt/C in HER region, (**c**) LSV curves after different stability cycles and (**d**) LSV curves at different bending angles.

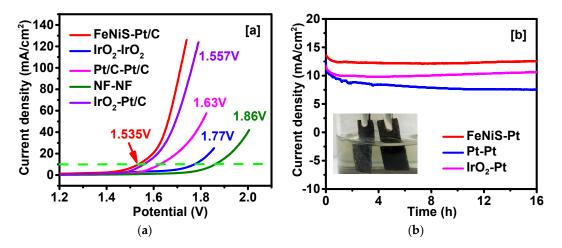


Figure 5. Results for overall water splitting using two electrode system: (**a**) polarization curves for FeNiS-Pt/C, IrO₂-IrO₂, Pt/C-Pt/C, NF-NF and IrO₂- Pt/C and (**b**) chronoamperometry plots for FeNiS-Pt/C, Pt/C-Pt/C and IrO₂-Pt/C.

The charge storage behavior of the synthesized samples was analyzed using cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) test. The relationship of the specific capacitance at different scan rates can be observed in Figure 6a. The highest specific capacitance of 13.2 F/cm² was observed for the FeNiS sample at a scan rate of 1 mV/s. It can be clearly seen from the CV curves in Figure 6b that FeNiS showed the highest area under CV curves at 2 mV/s compared to FeNiO and NF, leading to higher capacitance values. The capacitance increases with decreasing scan rates, which could be due to sufficient time provided for the faradic reaction representing the pseudocapacitive behavior (Figure S5a) [43,44]. The results of the galvanostatic charge-discharge test are represented by the relationship between specific capacitance with respect to the current density in Figure 6c. The discharge time for the FeNiS electrode was significantly higher compared to FeNiO and NF, as summarized in Figure 6d. As the charging current decreases, an improvement is observed in the charging and discharging time of the FeNiS sample (Figure S5b). In comparison, Li, Y. et al. observed 78.5% capacitance retention after 3000 cycles for Ni-Co sulfides [45]. In our study, the FeNiS showed a performance with 77.3% retention after 5000 cycles in charge-discharge measurements and comparable capacitance results to several other reports (Table S4 and Figure S5c).

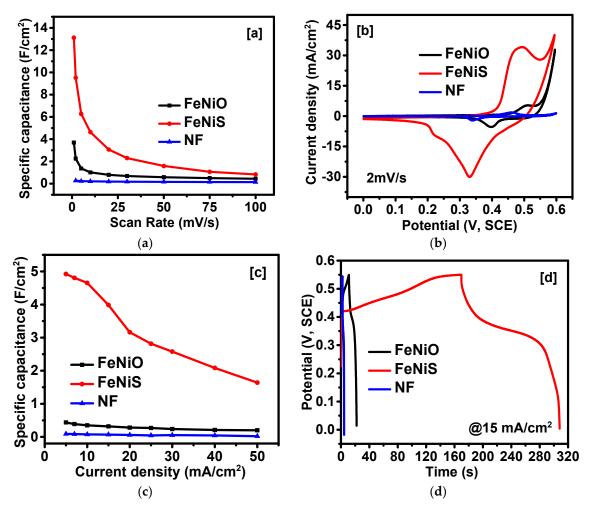


Figure 6. Charge storage behavior for FeNiS, FeNiO and NF: (**a**) specific capacitance vs. scan rate, (**b**) CV curves, (**c**) specific capacitance vs. current density and (**d**) charge discharge curves.

3. Experimental Details

3.1. Preparation of FeNiO and FeNiS Electrodes

In a typical process, 3 mmol of ammonium fluoride, 7.5 mmol of urea, 1 mmol of Ni(NO₃)₂·6H₂O and Fe(NO₃)₃·9H₂O were dissolved in 30 mL DI-water. The solution was then transferred into a 50-mL Teflon-lined autoclave reactor. The precleaned Ni foam (NF) was immersed in this solution, and the reaction was carried out at 120 °C for 6 h. The obtained NF was washed several times with water and ethanol. Similarly, the precipitates were washed and collected using a centrifuge at 10,000 rpm for 10 min. Resultant samples were dried at 60 °C for 12 h under vacuum. Further, FeNiO was obtained by calcining the samples at 350 °C for 2 h at a heating rate of 5 °C/min.

For the preparation of FeNiS, 500 mg thiourea was dissolved in 30 mL ethanol. The solution was transferred into a 50 mL Teflon-line autoclave reactor containing the non-calcined samples from the previous reaction. The reaction was carried out at 140 °C for 12 h. The obtained NF coated with FeNiS and the precipitates were washed and dried under vacuum. The comparison was carried out using commercial samples of IrO₂ and Pt-C coated over NF.

3.2. Characterization

A Shimadzu X-ray diffractometer using $CuK\alpha_1$ ($\lambda = 1.5406$ Å) radiation was used to study the crystal structure of the synthesized samples. The microstructural/morphological features of the FeNiO and FeNiS samples, hydrothermally deposited on Ni foams, were studied using a Quanta 200 SEM (FEI, Hillsboro, OR, USA) at an acceleration voltage of 20 kV. XPS analysis was performed over the scrap-out powder from the Ni foam electrode using a ThermoFisher Scientific Instruments K-Alpha⁺ spectrometer (East Grinstead, UK) to study the chemical composition of the synthesized samples. The XPS instrument was equipped with a monochromated Al K α radiation source (h ν = 1486.6 eV) and an X-ray spot of ~400 µm in radius. Survey spectra were acquired with a pass energy of 200 eV, while high-resolution core level spectra for all elements were acquired with a pass energy of 50 eV. The electrochemical characterization studies of the electrodes were carried out using a standard three-electrode system and a two-electrode system using a Versastat 4-500 electrochemical workstation (Princeton Applied Research, Oak Ridge, TN, USA). In the three-electrode system, synthesized samples were used as the working electrodes, a saturated calomel electrode (SCE) was used as the reference electrode and a graphitic rod was used as the counter electrode. The electrocatalytic behavior of the synthesized electrodes was measured in 1 M KOH electrolyte using linear sweep voltammetry, cyclic voltammetry, chronoamperometry (CA) and electrochemical impedance spectroscopy (EIS). The charge storage behavior of the electrodes was measured in 3 M KOH using a CV, GCD and EIS. The EIS measurements were performed in a frequency range of 0.05 Hz to 10 kHz, with an applied AC amplitude of 10 mV. The average weight loading of the material in all the electrodes was 1.2 mg/cm².

4. Conclusions

In conclusion, we have synthesized nano-flower shaped FeNiS on Ni foam using a facile hydrothermal method as a multifunctional electrode material for energy conversion and storage devices. The FeNiS electrode exhibited excellent stability and flexibility along with higher catalytic activity providing low overpotential values for the OER and HER reactions. Moreover, it showed competitive performance when used as the OER electrode in an electrolyzer device study over the standard benchmark IrO₂ system. Although competing with Pt/C as HER catalyst still remains a challenge, FeNiS showed a promising HER activity. The FeNiS electrode displayed acceptable charge storage capabilities to be used for supercapacitor materials. A high areal capacitance of 13.2 F/cm² was obtained at a scan rate of 1 mV/s. From our study, we report a cost-effective multifunctional electrode material for energy conversion and storage devices.

Supplementary Materials: The following are available online at http://www.mdpi.com/2073-4344/9/7/597/s1, Figure S1: XPS survey spectra of (a) FeNiO and (b) sulfurized FeNiO. Figure S2: (a) Nyquist plot for all the OER catalytic electrodes at 0.5 V, (b) Nyquist plot at different stability cycles at 0.5 V, (c) Nyquist plot at different bending angles at 0.5 V, and (d) chronoamperometry. Figure S3: (a) Nyquist plot at different stability cycles at -1.3 V, (b) Nyquist plot at different bending angles at -1.3 V and, (c) chronoamperometry. Figure S4: LSV polarization curves for Pt/C and IrO₂: (a) HER and (b) OER. Figure S5: Characteristics of FeNiS: (a) CV curves, (b) GCD curves and (c) cyclic stability plot for FeNiS sample using GCD measurements. Table S1. Comparison of OER catalysis with other reports. Table S2. Comparison of supercapacitor performance of some recently reported results.

Author Contributions: R.K.G. conceived the project, designed the experiments, interpreted the data, and finalized the manuscript. The first draft of the manuscript was written by C.Z. (Chen Zhao), C.Z. (Chunyang Zhang) and S.B. C.Z. (Chen Zhao), C.Z. (Chunyang Zhang) and S.B. performed all the electrochemical measurements, analyzed the data and recorded the XRD patterns. SEM images were recorded by G.C. M.B. and S.H. provided XPS data. P.K.K., N.K., C.M., C.R., P.K.K. and K.P. reviewed and finalized the manuscript.

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Conflicts of Interest: The authors declare no conflict of interest.

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